Nitrogen-Doped Graphene as Efficient Metal-Free Electrocatalyst for Oxygen Reduction in Fuel Cells

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latinum (Pt) nanoparticles have long been regarded as the best catalyst for the oxygen reduction reaction (ORR) in fuel cells, though the Pt-based electrode still suffers from its susceptibility to time-dependent drift and CO deactivation.1−3 Furthermore, the high cost of the Pt catalysts, together with the limited reserves of Pt in nature, has been shown to be the major “showstopper” to mass market fuel cells for commercial applications. That is why the large-scale practical application of fuel cells has not been realized, though alkaline fuel cells with platinum as an ORR electrocatalyst were developed for the Apollo lunar mission in the 1960s.7 Along with recent intensive research efforts in reducing or replacing Pt-based electrodes in fuel cells,5,11 we have found that vertically aligned nitrogen-containing carbon nanotubes (VA-NCNTs) produced by pyrolysis of iron(II) phthalocyanine (a metal heterocyclic molecule containing nitrogen),12 in either the presence or absence of additional NH3 vapor,3 could act as effective metal-free ORR electrocatalysts. The metal-free VA-NCNTs were shown to catalyze a four-electron ORR process free from CO “poisoning” with a 3-time higher electrocatalytic activity, smaller crossover effect, and better long-term operation stability than that of commercially available Pt-based electrodes (C2−20, 20% platinum on VulcanXC-72R; E-TEK) in alkaline electrolytes.3 On the basis of the experimental observations and quantum mechanics calculations, we attributed the improved catalytic performance to the electron-accepting ability of the nitrogen atoms, which creates a net positive charge on adjacent carbon atoms in the nanotube carbon plane of VA-NCNTs to readily attract electrons from the anode for facilitating the ORR.3 Uncovering this new ORR mechanism in nitrogen-doped carbon nanotube electrodes is significant as the same principle could be applied to the development of various other metal-free efficient ORR catalysts for fuel cell applications.

The recent discovery of graphene has opened up a new era of 2-dimensional (2D) fundamental science and potential technology.13−15 As mother of all graphitic forms, graphene is a building block for carbon materials of all other dimensionalities, such as 0D buckyballs, 1D nanotubes, and 3D graphite. Having many similarities to carbon nanotubes (CNTs) in structure and property, including its high aspect ratio (the ratio of lateral size to thickness), large surface, rich electronic states, and good mechanical properties, graphene is an attractive candidate for potential uses in many areas where the CNTs have been exploited. Superior to CNTs, the one atomic-thick graphene sheets with a 2D planar geometry will further facilitate electron transport,16 and hence the more effective electrode materials.

ABSTRACT Nitrogen-doped graphene (N-graphene) was synthesized by chemical vapor deposition of methane in the presence of ammonia. The resultant N-graphene was demonstrated to act as a metal-free electrode with a much better electrocatalytic activity, long-term operation stability, and tolerance to crossover effect than platinum for oxygen reduction via a four-electron pathway in alkaline fuel cells. To the best of our knowledge, this is the first report on the use of graphene and its derivatives as metal-free catalysts for oxygen reduction. The important role of N-doping to oxygen reduction reaction (ORR) can be applied to various carbon materials for the development of other metal-free efficient ORR catalysts for fuel cell applications, even new catalytic materials for applications beyond fuel cells.

KEYWORDS: graphene · N-doping · oxygen reduction · fuel cell

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Although CNTs and their N-doped counterparts have been synthesized and studied for some years, the large-scale preparation of graphene sheets by chemical vapor deposition (CVD) is only the recent development.\textsuperscript{17,18} More recently, attempts have been made to introduce heteroatoms (e.g., nitrogen) into graphene sheets for modulation of their electronic properties.\textsuperscript{19} As far as we are aware, however, the possibility for nitrogen-containing graphene sheets (N-graphene, we designate the N-free graphene as C-graphene for convenience) to be used as catalysts for the ORR process in fuel cells has not been exploited. In the present study, we investigated the CVD-deposited N-graphene sheets for ORR at the cathode in alkaline fuel cells. As we shall see later, the N-graphene showed a much better electrocatalytic activity, long-term operation stability, and tolerance to crossover and poison effects than the commercially available platinum-based electrodes (C2/H\textsubscript{11}002, 20% platinum on VulcanXC-72R; E-TEK) for oxygen reduction.

In this study, we used a modified CVD process\textsuperscript{17} for the synthesis of N-graphene films. Briefly, a thin layer of nickel (\textasciitilde 300 nm) was deposited on a SiO\textsubscript{2}/Si substrate by sputter coating. The Ni-coated SiO\textsubscript{2}/Si wafer was then heated up to 1000 °C within a quartz tube furnace under a high purity argon atmosphere. Thereafter, a nitrogen-containing reaction gas mixture (NH\textsubscript{3}: CH\textsubscript{4}:H\textsubscript{2}:Ar = 10:50:65:20 standard cubic centimeters per minute) was introduced into the quartz tube and kept flowing for 5 min, followed by purging with a flow of NH\textsubscript{3} and Ar only for another 5 min. The sample was then rapidly moved out from the furnace center (1000 °C) under Ar protection. The resultant N-graphene film can be readily etched off from the substrate by dissolving the residual Ni catalyst layer in an aqueous solution of HCl,\textsuperscript{17,18} allowing the free-standing N-graphene sheets to be transferred onto substrates suitable for subsequent investigation.

Figure 1a shows a free-standing film of ca. 4 cm\textsuperscript{2} in size floating on water after removing the nickel layer by dissolving in an aqueous acid solution; (b, c) AFM images of the N-graphene film and the corresponding height analyses along the lines marked in the AFM image (c1\textendash c3 in panel c).

Figure 1. (a) A digital photo image of a transparent N-graphene film floating on water after removal of the nickel layer by dissolving in an aqueous acid solution; (b, c) AFM images of the N-graphene film and the corresponding height analyses along the lines marked in the AFM image (c1\textendash c3 in panel c).

Figure 2. TEM and Raman analyses of the N-graphene films. (a) Low-magnification TEM image showing a few layers of the CVD-grown N-graphene film on a grid. Inset shows the corresponding electron diffraction pattern. (b\textendash d) High magnification TEM images showing edges of the N-graphene film regions consisting of (b) 2, (c) 4, and (d) ca. 4\textendash 8 graphene layers. (e) The corresponding Raman spectra of the N-graphene films of different graphene layers on a SiO\textsubscript{2}/Si substrate (Methods).

The substrate-free N-graphene sheets can be directly transferred onto TEM grids for further characterization. As shown in Figure 2a, TEM images also revealed flat flakes, resembling those seen in Figures 1b,c. Unlike most C-graphene films of symmetric hexagonal diffraction patterns,\textsuperscript{18,24} electron diffraction of the N-graphene film (inset of Figure 2a) shows a
ringlike diffraction pattern with dispersed bright spots. The above observed difference indicates that the otherwise crystalline graphene sheets became partially mis-oriented in the N-graphene film due to structure distortions caused by the intercalation of nitrogen atoms into its graphitic plans, as also seen for N-doped CNTs. The cross-sectional view of the suspended edge of the as-synthesized N-graphene film shows, once again, only a few layers (typically, 2−8 layers) of graphene sheets (Figures 2b–d). Adjacent interlayer distances in the N-graphene film were measured to be 0.3−0.4 nm, close to the d-spacing of (002) crystal plane (0.335 nm) of bulk graphite with slight distortion.

Raman spectra of the as-synthesized N-graphene sheets with different layer numbers are given in Figure 2e. Similar to C-graphene films,17,18,24 the G band and 2D band of the N-graphene samples were found to be sensitive to the layer number. For comparison, Raman spectra for both the N-graphene and C-graphene films were given in Supporting Information, Figure S2, which revealed a relatively higher D band for the N-graphene than that of C-graphene film due to structural distortion caused by N-doping. However, the very low value of 0.06−0.25 for the I_D/I_G (a peak intensity ratio of the D to G band)18 indicates that the N-graphene layers remain a high crystalline quality. This is further evidenced by the X-ray diffraction (XRD) profiles shown in Supporting Information, Figure S3, in which a pronounced (002) peak at 26° characteristic of C-graphene was also seen for N-graphene. The broad background band centered at 26° can be attributable to the presence of intercalated “N defects” in the N-graphene structure. The slight shift of the (002) peak from 26.5°, characteristic of C-graphene, to 26.1° for N-graphene seen in the inset of Supporting Information, Figure S3 suggests that the N-doping has also led to an increased interlayer spacing for the N-graphene film. From the (002) diffraction peak positions, we calculated the d-spacing for the N-graphene film to be 3.41 Å. This value is in good agreement with the TEM measurements, but still slightly higher than that of C-graphene film (3.36 Å) (Supporting Information, Figure S3) and graphite (3.35 Å).26

To probe nitrogen atoms in the N-graphene structure, we carried out X-ray photoelectron spectroscopic (XPS) measurements. As can be seen in Figure 3, the XPS survey spectrum for the N-graphene film shows a predominant narrow graphitic C 1s peak at 284.2 eV,27,28 along with a N 1s peak at ca.400 eV. Figure S4 shows the XPS survey spectra for both the N-graphene and C-graphene over a wide range of binding energies (0−1000 eV). In addition to the C 1s and N 1s peaks, an O 1s peak at ca. 540 eV was also observed for both N-graphene and C-graphene,29,30 possibly due to the incorporation physically adsorbed oxygen as in the case with CNTs.28,31 Thus, the higher O 1s peak relative to the corresponding C 1s peak seen for the N-graphene film shows, once again, only a few layers (typically, 2−8 layers) of graphene sheets (Figures 2b–d). Adjacent interlayer distances in the N-graphene film were measured to be 0.3−0.4 nm, close to the d-spacing of (002) crystal plane (0.335 nm) of bulk graphite with slight distortion.25

The above observed difference indicates that the otherwise crystalline graphene sheets became partially mis-oriented in the N-graphene film due to structure distortions caused by the intercalation of nitrogen atoms into its graphitic plans, as also seen for N-doped CNTs.32−35 Both Raman and XRD results suggested that the N-graphene film prepared under the same CVD conditions, but without the introduction of NH3 gas, and commercially available Pt-loaded carbon (Vulcan XC-72R) supported by a glassy carbon electrode (Pt/C) were also investigated for comparison. Just like the pure CNT electrode without N-doping,3 the C-graphene electrode shows a two-step, four-electron process for oxygen reduction with a low onset potential of about −0.45 V (Figure 4a). Unlike the C-graphene electrode, the N-graphene electrode (Figure 4a) exhibited a one-step, four-electron pathway for the ORR, as is the case with N-doped CNTs3. The steadystate catalytic current density at the N-graphene electrode was found to be ca. 3 times higher than that of the Pt/C electrode over a large potential range (Figure 4a). The transferred electron number (n) per oxygen molecule at the N-graphene electrode was calculated by Koutecky−Levich (K−L) equation27 to be 3.6−4 at the potential ranging from −0.4 to −0.8 V (Supporting Information, Figure S5). These results indicate that the N-graphene electrode,
similar to its CNT counterpart,\textsuperscript{3} is a promising metal-free catalyst for the ORR in an alkaline solution. The N-graphene electrode was further exposed to fuel molecules (e.g., methanol) and CO to test the possible crossover and poison effects.\textsuperscript{38,39} We measured electrocatalytic selectivity of the N-graphene electrode against the electro-oxidation of various commonly used fuel molecules, including hydrogen gas, glucose, and methanol (Figure 4b). Also included in Figure 4b is the corresponding current density ($j$)–time ($t$) chronoamperometric response at a Pt/C electrode for comparison. As shown in Figure 4b, a 40% decrease in current appears at the Pt/C electrode upon the addition of 2% (w/w) methanol into the air-saturated electrochemical cell. Current ($j$)–time ($t$) chronoamperometric response of Pt/C (circle line) and N-graphene (square line) electrodes to CO. The arrow indicates the addition of 10% (v/v) CO into air saturated 0.1 M KOH at $-0.4$ V; $j_o$ defines the initial current. (d) Cyclic voltammograms of N-graphene electrode in air saturated 0.1 M KOH before (circle line) and after (square line) a continuous potentiodynamic swept for 200000 cycles at room temperature (25 °C). Scan rate: 0.1 V/s.

Figure 4. (a) RRDE voltammograms for the ORR in air-saturated 0.1 M KOH at the C-graphene electrode (red line), Pt/C electrode (green line), and N-graphene electrode (blue line). Electrode rotating rate: 1000 rpm. Scan rate: 0.01 V/s. Mass$_{\text{Pt/C}}$/Mass$_{\text{N-graphene}}$ = 7.5 µg. (b) Current density ($j$)–time ($t$) chronoamperometric responses obtained at the Pt/C (circle line) and N-graphene (square line) electrodes at $-0.4$ V in air saturated 0.1 M KOH. The arrow indicates the addition of 2% (w/w) methanol into the air-saturated electrochemical cell. (c) Current ($j$)–time ($t$) chronoamperometric response of Pt/C (circle line) and N-graphene (square line) electrodes to CO. The arrow indicates the addition of 10% (v/v) CO into air saturated 0.1 M KOH at $-0.4$ V; $j_o$ defines the initial current. (d) Cyclic voltammograms of N-graphene electrode in air saturated 0.1 M KOH before (circle line) and after (square line) a continuous potentiodynamic swept for 200000 cycles at room temperature (25 °C). Scan rate: 0.1 V/s.

In summary, we have developed a facile CVD approach for the large-area synthesis of N-graphene films, which can be readily transferred onto various substrates. The high-quality N-graphene films thus formed were evidenced by AFM, TEM, Raman, XRD, and XPS measurements. The N-graphene film shows superb performance for oxygen reduction reaction associated with alkaline fuel cells. The steady-state catalytic current at the N-graphene electrode was found to be $ca$. 3 times higher than that at the Pt/C electrode over a large potential range. The long-term stability, tolerance to crossover, and poison effect are also better than Pt/C for oxygen reduction in alkaline electrolyte. It is notable that the N-doped graphene film possesses remarkable electrocatalytic properties for ORR similar to that of nitrogen-containing vertically aligned carbon nanotubes.\textsuperscript{3} The ease with which graphene materials and their N-doped derivatives can be produced by various low-cost large-scale methods, including the chemical vapor deposition, chemical reduction of graphite oxide,\textsuperscript{14,20} exfoliation of graphite,\textsuperscript{21} microwave plasma reaction,\textsuperscript{41} and atmospheric pressure graphitization of silicon carbide,\textsuperscript{42} suggests considerable room for cost-effective preparation of metal-free efficient graphene-
based catalysts for oxygen reduction. Furthermore, this work clearly indicates that the important role of N-doping to ORR demonstrated with nitrogen-doped carbon nanotubes and N-graphene can be applied to other carbon materials for the development of various other metal-free efficient ORR catalysts for fuel cell applications, even new catalytic materials for applications beyond fuel cells.

METHODS

Preparation of N-Graphene Films. A thin layer of nickel film (~300 nm) was deposited on a SiO2/Si substrate by sputter coating, which was then heated up to 1000 °C within a quartz tube under a high purity argon atmosphere. Thereafter, a nitrogen-containing reaction gas mixture (NH3:CH4:H2:Ar = 10:50:65:200 standard cubic centimeters per minute) was introduced into the quartz tube and kept flowing for 5 min, followed by purging with a flow of NH3 and Ar only for another 5 min. The sample was then rapidly moved out of the furnace center (1000 °C) under Ar protection. The resultant N-containing graphene (designated as: N-graphene) film can be readily etched off from the substrate by dissolving the Ni catalyst residues in an aqueous solution of HCl (1 M), allowing the free-standing N-graphene sheets to be transferred onto substrates suitable for subsequent investigation. For comparison, the pure graphene film was synthesized by the same procedure without NH3 flow.17

Electrode Preparation. N-graphene and pure graphene (designated as C-graphene) nanofilms floating on DI water were transferred onto a glassy carbon (GC) disk electrode (3 mm in diameter, Pine Research Instrumentation). The electrode was dried at 40 °C under vacuum for 60 min. Electrodes for rotating ring-disk electrode (RRDE) voltamogram measurements were prepared on a 5 mm diameter GC disk electrode (Pine Research Instrumentation). The commercialized Platinum loaded carbon catalysts (Pt/C, C20–20, 20% Pt on Vulcan XC-72R, E-TEK Division, PEMAS Fuel Cell Technologies) electrode was prepared according to the procedure described elsewhere.1 Typically, 1 mg/ml PtC dispersion in ethanol was prepared using ultrasonication. A 7.5 µg portion of PtC was introduced onto the GC electrode using a microsyringe.

Characterization. Scanning electron microscopy (SEM) and transmission electron microscopy (TEM) images were recorded on a Hitachi S4800-F high-resolution SEM and H-7600 TEM unit (Hitachi, Japan), respectively. Tapping-mode atomic force microscopy (AFM) measurements were preformed on Nano-R2 SPM system (Pacific Nanotechnology). X-ray photoelectron spectroscopic (XPS) measurements were performed on a VG Microtech ESCA 2000 using a monochromatic Al Kα x-ray source. X-ray diffraction patterns of C-graphene and N-graphene films were performed on a Rigaku Miniflex II diffractometer with Cu Kα radiation (1.406 Å). The diffraction data were collected in step scans, with a step size of 0.05° (2θ) and a count time of 2 s per step between 10 and 100° (2θ). Raman microspectroscopy was carried out on a Renishaw inVia unit using the Ar ion laser with an excitation wavelength of 514.5 nm.

Electrochemical measurements were performed on a computer-controlled potentiostat (CHI 760C, CH Instrument, USA) with a three-electrode cell equipped with gas flow systems. A N-graphene, C-graphene, or Pt/C electrode was used as working electrode while an Ag/AgCl (4 M KCl-filled) electrode as reference and a platinum wire as counterelectrode.

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Supporting Information Available: Additional SEM image, Raman, XRD, XPS, and J-V curves for N-graphene and C-graphene are given in Figures S1–S5. This material is available free of charge via the Internet at http://pubs.acs.org.

REFERENCES AND NOTES